

## Synthesis and Characterization of Samarium doped $\text{SrBi}_{2-x}\text{O}_4$ oxides

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**Abstract:**  $\text{Sm}^{3+}$  doped  $\text{SrBi}_{2-x}\text{O}_4$  oxides were prepared by sol - gel method via nitrate citrate route with 'x' values ranging from 0.0, 0.1, 0.2, 0.3, and 0.4 (samples C0 - C4). X-ray powder diffraction analysis reveals the single phase monoclinic structure with the space group of  $P\bar{2}m$ . Rietveld refinement of unit cell structure shows lowered agreement factors  $R_p$ ,  $R_{wp}$  and  $R_{exp}$  for all the samples. Average crystallite size was determined by Scherrer's relation in the range of ~36 - 45 nm. Magnetic properties of the samples were analyzed by using Vibrating Sample Magnetometer at room temperature which reveals a weak ferromagnetic behaviour for C1 to C4 and diamagnetic behaviour for sample C0. EPR studies recorded at liquid helium temperature show a single absorption peak with defect line content. Optical absorption spectra of the prepared samples were recorded in UV visible region from 200 nm to 800 nm. FT-IR spectra of the samples were recorded at room temperature, which show the characteristic absorptions at  $853 \text{ cm}^{-1}$  and  $533 \text{ cm}^{-1}$  corresponding to the tetrahedral and octahedral sites. SEM studies show the particles were agglomerated with rod like structure. The combined TGA – DSC measurements show important stages of decomposition with no phase transition.

**Keywords:** Powder X-ray diffraction, Samarium ion, TGA, FT-IR, Sol – gel synthesis.

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### I. Introduction

Growing demand for new research material with practical application necessity is growing everyday in view of scientific and technological interests. If a material is magnetically active nano metal oxide, it receives much attention owing to their application in memory storage device, data writing, gas sensors and in magnetic drug delivery system [1-2]. Choosing suitable magnetically active ion with excellent electrical and optical properties, one can cater the needs of growing demands. Rare earth ion with unpaired  $4f$  electron when doped in to larger size cation leads to significant structural distortion with strain and modifies the electrical and magnetic properties [3-6]. The method of preparation of this composite oxide material should involve low temperature, homogenous particles in nano range and good stoichiometric control. Several soft methods are available for the synthesis of magnetically active composite oxide material such as hydrothermal, micro emulsion, spray drying, sol – gel and reverse micelle method. Among these methods the sol – gel method [7] is one of the best method for preparing magnetically active composite oxide material which has the ability to provide all the desired properties.

### II. Experimental

The sol-gel method was used for the synthesis of  $\text{Sm}^{3+}$  doped  $\text{SrBi}_{2-x}\text{O}_4$  oxides.  $\text{Sm}^{3+}$  doped samples were synthesized by mixing  $\text{SrCO}_3$ ,  $\text{Sm}_2\text{O}_3$  and  $\text{Bi}(\text{NO}_3)_2$  in a stoichiometric ratio to obtain  $\text{SrSm}_x\text{Bi}_{2-x}\text{O}_4$  oxides with x values from 0, 0.1, 0.2, 0.3, 0.4. Starting materials  $\text{SrCO}_3$ ,  $\text{Sm}_2\text{O}_3$  and  $\text{Bi}(\text{NO}_3)_2$  were first mixed with aqueous  $\text{HNO}_3$  solution ( $\text{pH} \sim 2$ ) for converting into corresponding nitrates. In addition to nitric acid, urea was also added to  $\text{Sm}_2\text{O}_3$  solution to act as a trigger during combustion process. The above said solution was mixed with 30 ml of 1.5 M citric acid and kept in magnetic stirrer at  $50^\circ\text{C}$  for four days. The resulted gel like substance was then decomposed at  $120^\circ\text{C}$  and sintered at  $400^\circ\text{C}$  for 2 hours and then at  $800^\circ\text{C}$  for 4 hours in a muffle furnace. The resulting powder obtained was ground in an agate mortar and utilized for characterization.

#### 2.1. Characterization

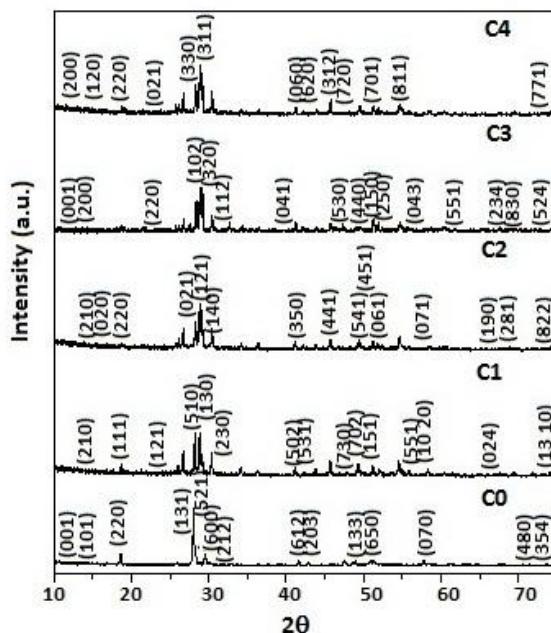
X-ray powder diffraction studies were carried out in an X'pert powder X-ray diffractometer (PAN Analytical make) with a scan rate  $27^\circ$  / minute in the range of  $10^\circ$  –  $75^\circ$  in  $2\theta$ . Monochromatic  $\text{Cu K}_\alpha$  radiation ( $\lambda \sim 1.54060 \text{ \AA}$ ) was used as X-ray source with a power 40 kV/30mA. Rietveld refinement method was used for the refinement of unit cell structure. FT-IR spectra are recorded using Thermo Scientific FT-IR spectrometer in the range  $4000$  –  $450 \text{ cm}^{-1}$ . Surface structure and particle morphology was studied using SEM (JSM – 5410). Optical absorption spectrum was recorded in a UV – Vis spectral region of 200 – 800 nm (UV 2450 Make: Shimadzu). Hitchi S – 3400 instrument was used for elemental analysis. Magnetic hysteresis measurements were made by using Vibrating Sample Magnetometer at room temperature (Make: Lake Shore, Model: 7404).

EPR spectra were recorded in a Bruker spectrometer in the X Band at Liquid helium with a temperature range of 6 – 70K. TGA measurement were recorded by using TA instruments (Model: Q600 and Q20 DSC). Density measurements were done by using liquid displacement method using Carbon tetrachloride as an immersion liquid (density 1.596 g/cc at 300K)

### III. Results And Discussion

#### 3.1. X – ray diffraction studies

XRD patterns of prepared samples with different doping amount of  $\text{Sm}^{3+}$  ions are given in figure-1. Sharp diffraction peaks are observed for XRD pattern of C0 to C4. Multiplets of peaks from C1 to C4 indicate the successful incorporation of  $\text{Sm}^{3+}$  ion. The doping of  $\text{Sm}^{3+}$  ion ( $0.964\text{ \AA}$ ) which has smaller ionic radius than  $\text{Bi}^{3+}$  ( $1.03\text{ \AA}$ ), as a result it leads to structural distortion in the lattice sites [8-10]. Cell volume increases for concentration C0 to C2 and decline for C3 and C4. Average crystallite size is calculated by Scherrer formula and found to be in the range of 36 to 45nm. Though the compositional formula of  $\text{SrBi}_2\text{O}_4$  represents spinel oxide formula  $\text{AB}_2\text{O}_4$ ,  $\text{SrBi}_2\text{O}_4$  is a non spinel oxide which crystallizes in monoclinic state as reported earlier [11-12]. Structural investigations using standard Rietveld Profile refinement indicate monoclinic phase with no other secondary phase having a space group of P / 2m, which is in agreement with the respective joint committee on powder diffraction standards. The agreement factors after Rietveld refinement of unit cell structure of  $\text{SrSm}_x\text{Bi}_{2-x}\text{O}_4$  are given in the table 1.



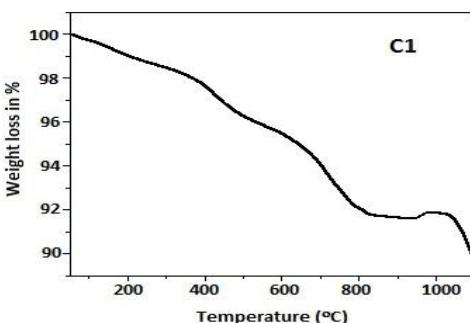
**Fig. 1.** Powder X-ray diffraction patterns of  $\text{Sm}^{3+}$  substituted  $\text{SrBi}_{2-x}\text{O}_4$  samples C0 to C4

Agreement Factors			
Sample ID	$R_p$ (%)	$R_{wp}$ (%)	$R_{exp}$ (%)
C0	100.63	98.82	0.06
C1	95.79	94.38	0.07
C2	97.14	97.18	0.08
C3	91.92	99.88	0.09
C4	97.42	96.17	0.06

**Table. 1.** Agreement factors after Rietveld refinement of  $\text{SrSm}_x\text{Bi}_{2-x}\text{O}_4$  unit cell structure

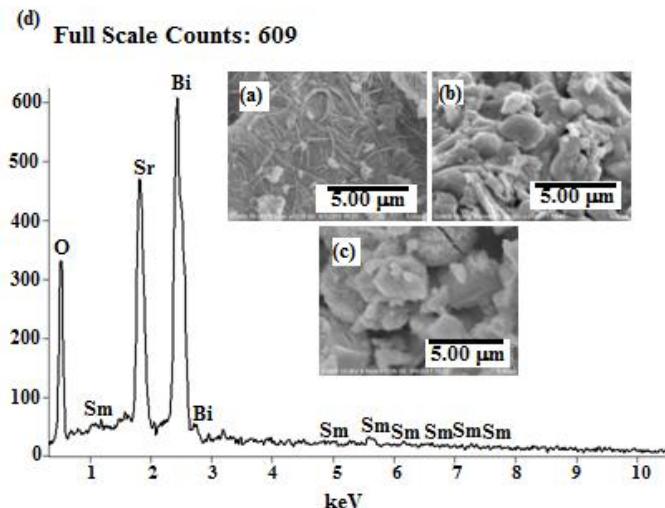
#### 3.2. Thermal analysis

From TG curve, it is evident that weight loss starts gradually from room temperature (Fig. 2). First stage of decomposition starts from room temperature to  $400^\circ\text{C}$  which might be due to elimination of trapped water molecule. In general sol gel method shows weight loss due to dehydration by 5% [13]. In second stage, there is a slight weight loss at  $400^\circ\text{C}$  which is due to removal of oxygen present in the sample. To conclude there is no significant thermal change and the compound is thermally stable up to  $800^\circ\text{C}$ .

**Fig. 2.** TG curve of  $\text{SrBi}_2\text{O}_4:\text{Sm}^{3+}$  for  $x = 0.15$ 

### 3.3. SEM and Elemental analysis

Scanning electron micrograph and EDX of prepared samples are shown in the figure-3. Nature of  $\text{Sm}^{3+}$  doped  $\text{SrBi}_{2-x}\text{O}_4$  oxide shows nano rod formation for C0. From C1 to C4 agglomeration can be seen for the prepared samples with the increase in concentration of  $\text{Sm}^{3+}$  ion. Incorporation of samarium ion in the lattice reduces the pores and void present in the lattice structure. Elemental analysis confirms the presence of oxygen, samarium, strontium and bismuth in the stoichiometric proportion (Table 2). Elemental analysis results show that the prepared samples are free from impurity.

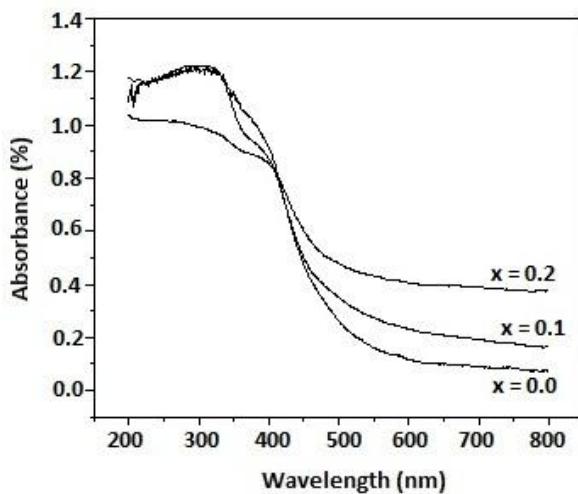
**Fig.3.** SEM images of (a) C0 (b) C1 (c) C4 at different magnification and (d) EDX profile of C1 with corresponding elemental composition values

Sample ID	Sr (wt %)	Sm (wt %)	Bi (wt %)	O (wt %)
C0	12.05	-	69.62	18.34
C1	17.94	3.77	49.13	29.15
C2	16.75	6.70	50.40	26.15
C3	16.21	12.08	46.91	24.80
C4	18.54	11.32	45.35	24.79

**Table. 2.** Elemental composition of prepared samples using EDX

### 3.4. UV-Visible absorption studies

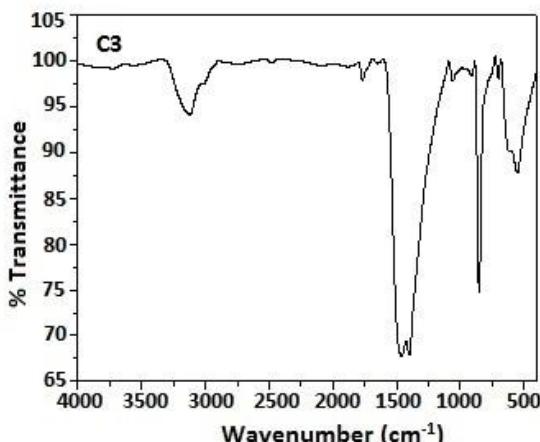
UV-Vis absorption spectra of  $\text{Sm}^{3+}$  doped  $\text{SrBi}_{2-x}\text{O}_4$  oxides for composition  $x = 0, 0.1$  and  $0.2$  is shown in figure-4.  $\text{Sm}^{3+}$  doped  $\text{SrBi}_{2-x}\text{O}_4$  oxides shows optical absorption in the range  $350 - 400\text{nm}$ . As the concentration of  $\text{Sm}^{3+}$  increases, absorption shows a mild shift to longer wavelength. This is due to the charge transfer from  $2p$  orbital of oxygen atom to  $4f$  orbital of  $\text{Sm}^{3+}$  ion. Undoped  $\text{SrBi}_{2-x}\text{O}_4$  shows absorption at  $340\text{nm}$  and substitution of  $\text{Sm}^{3+}$  increases the absorption wavelength to  $410\text{nm}$ .



**Fig. 4.** UV-Vis absorption spectra of  $\text{Sm}^{3+}$  doped  $\text{SrBi}_{2-x}\text{O}_4$  oxides

### 3.5. Fourier Transform Infrared

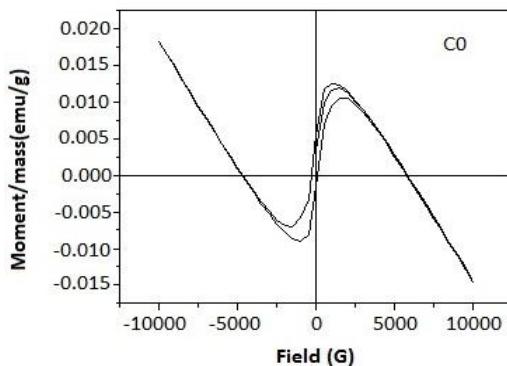
A representative FT-IR spectrum is given in the figure-5 for sample C3. Broad band in the range  $3124\text{ cm}^{-1}$  is assigned to stretching mode of OH group and absorption in the range  $1766\text{ cm}^{-1}$  assigned to bending mode of OH group. Vibration around  $1470\text{ cm}^{-1}$  to  $1390\text{ cm}^{-1}$  assigned to antisymmetric NO stretching vibration because of residual nitrate present in the sample [14]. In the region less than  $1000\text{ cm}^{-1}$  there are absorption band at  $906\text{ cm}^{-1}$ ,  $853\text{ cm}^{-1}$ ,  $703\text{ cm}^{-1}$ ,  $629\text{ cm}^{-1}$  and  $553\text{ cm}^{-1}$ . Absorption at  $533$  and  $629\text{ cm}^{-1}$  is due to Bi-O stretching vibration at  $[\text{BiO}_6]$  units at octahedral sites and weak shoulder at  $703\text{ cm}^{-1}$  corresponds to symmetric stretching vibration of Bi-O bond in pyramidal  $[\text{BiO}_3]$  units [14,15]. Broad band at  $853\text{ cm}^{-1}$  is assigned to Sr-O bond [12] or due to vibration of  $\text{Sm}^{3+}-\text{O}^{2-}$  bond [16].



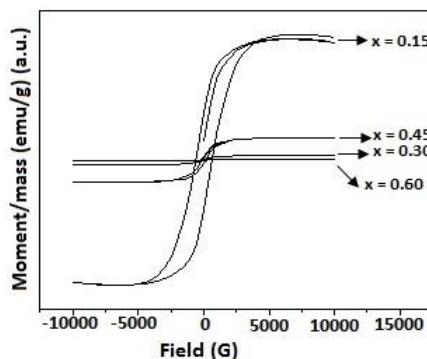
**Fig. 5.** FT-IR Spectra  $\text{SrSm}_x\text{Bi}_{2-x}\text{O}_4$  of sample C3 at room temperature

### 3.6. Magnetic measurements

Magnetic hysteresis loops of  $\text{Sm}^{3+}$  doped  $\text{SrBi}_{2-x}\text{O}_4$  oxides C1 to C4 show a soft ferromagnetic behaviour at room temperature (Fig. 6 & 7). Sample C0 exhibit diamagnetic behaviour owing to the absence of  $\text{Sm}^{3+}$  ion as  $\text{Sr}^{2+}$  and  $\text{Bi}^{3+}$  shows diamagnetic character. Samples C1 to C4 show a weak ferromagnetic behaviour in the order of  $10^{-3}$  emu/g which indicates successful incorporation of  $\text{Sm}^{3+}$  ion (Table 3). Paramagnetic nature of  $\text{Sm}^{3+}$  is mainly due to the presence of unpaired  $4f$  electron. From VSM figure, one can infer that ferromagnetic curve for  $x = 0.30$  and  $0.60$  are very narrow which is due to low values of saturation magnetization  $M_s$  and magnetic remanence  $M_r$ . Increase in samarium concentration leads to the porosity which may be the reason for the declining  $M_s$  values. Also variation in  $M_s$  values are due to the different cation distribution and surface structural distortion. Variation in coercivity values are attributed to movement of domain walls in response to the magnetic field [17].



**Fig. 6.** Hysteresis loop showing diamagnetic behaviour of  $\text{SrBi}_2\text{O}_4$



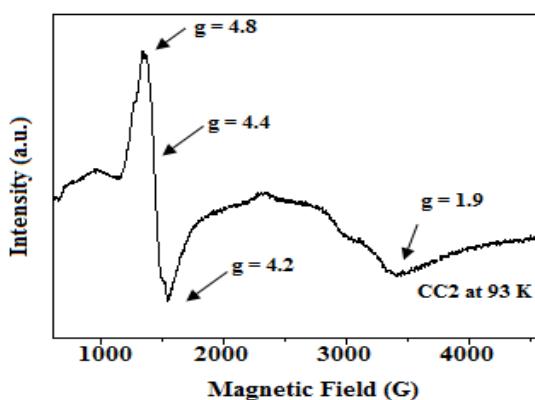
**Fig.7.** Hysteresis loop showing ferromagnetic behaviour of  $\text{SrSm}_x\text{B}_{2-x}\text{O}_4$  series

Sample ID	Magnetic Properties		
	$M_s$ (emu/g)	$M_r$ (emu/g)	$H_c$ (G)
C0	$16.446 \times 10^{-3}$	$3.527 \times 10^{-3}$	3069.5
C1	$18.908 \times 10^{-3}$	$7.342 \times 10^{-3}$	597.37
C2	$0.754 \times 10^{-3}$	$0.050 \times 10^{-3}$	70.073
C3	$3.366 \times 10^{-3}$	$0.523 \times 10^{-3}$	151.85
C4	$0.147 \times 10^{-3}$	$0.013 \times 10^{-3}$	83.332

**Table 3.** Magnetisation values of  $\text{SrSm}_x\text{B}_{2-x}\text{O}_4$

### 3.7. EPR studies

Figure-5 shows EPR spectra of  $\text{Sm}^{3+}$  doped  $\text{SrBi}_2\text{O}_4$  oxide with  $x = 0.30$  at 93 K. Electronic configuration of  $\text{Sm}^{3+}$  is  $4f^5$ . Rare earth ion with half filled  $4f$  shell is strongly coupled to lattice fluctuations except  $\text{Gd}^{3+}$  ion with  $4f^5$  configuration. Due to this, these ions have short spin-lattice relaxation times and as a result, EPR spectra can be observed only at low liquid helium temperature. From the EPR spectra, we can observe two absorption signals. First absorption signal at  $g \sim 1.9$  is weak, less intense and broadened due to unresolved hyperfine splitting which indicate the presence of paramagnetic site due to  $\text{Sm}^{3+}$  ion [18-20]. Second absorption signal at  $g \sim 4.2$  to 4.8 is broad, unresolved giving a long shoulder like structure.



**Fig.7.** EPR Spectra of  $\text{Sm}^{3+}$  doped  $\text{SrSm}_x\text{B}_{2-x}\text{O}_4$  oxide at Liquid helium temperature when  $x = 0.2$

#### IV. Conclusion

Sm<sup>3+</sup> doped SrBi<sub>2</sub>O<sub>4</sub> oxides with varying composition were synthesized by sol gel method. Powder XRD analysis confirms the presence of monoclinic structure with space group P / 2m. TGA-DSC reveals no significant weight loss and all the samples are thermally stable up to 800 °C. FT-IR analysis confirms the stretching vibration of Bi-O bond in octahedral coordination sites. EPR studies reveal weak paramagnetic absorption with g ~1.9 with defect line content. Scanning electron microscope shows grains which are agglomerated with irregular particle size. UV-Vis absorption studies shows absorption in the range 350-400nm. Magnetic behaviour of samples shows weak ferromagnetic of the order of 10<sup>-3</sup> emu/g. Decrease in magnetic values is due to increase in samarium concentration which reduces magnetic behaviour between strontium and bismuth sites.

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