## Augmentation Of The Supercapacitive Performance Of Mn<sub>3</sub>O<sub>4</sub> Nanoparticles By Doping Ni.

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#### Abstract:

Ni-doped  $Mn_3O_4$  nanoparticles (NMO) with a morphology facilitating high surface area were synthesized by prudent hydrothermal synthesis, manifesting a supercapacitive tone with high specific capacitance and excelling cyclability. NMO nanoparticles were agglomerated from hydrated manganese chloride, hydrated nickel chloride, and sodium hydroxide precursors. The XRD results demonstrated predominant peaks oriented from (211), analogous to the 141/amd space group tetragonal structure. The Raman studies affirmed the bonding characteristics of vibration and bending modes of the Mn-O bond. The SEM inquiries proclaimed the morphology providing a large surface area. EDAX reports infer the elemental composition for phase purity confirmation of elements Mn, O, and Ni. Electrochemical accusation exhibited a pseudocapacitive character through cyclic voltammetry. The GCD deductions evince a specific capacitance of 565 F/g for  $Mn_3O_4$ electrodes and 626 F/g for 0.6% NMO electrodes at a specific current of 0.5 A/g. These NMO electrodes achieved a cyclic capacitance rate of 82% after 3600 cycles.

Key Words: Hydrothermal synthesis, Ni:Mn<sub>3</sub>O<sub>4</sub> nanoparticles and specific capacitance.

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#### I. Introduction

The world lay claim to the burgeoning of electrochemical storage devices for copious storage of electrical energy [1-3]. Stereotype of devices are supercapacitors. Supercapacitors (SC) or ensorcelled electrochemical capacitors, have array of applications in electric, electronic and communications widgets [4]. In low-energy-density and high-power-density applications, batteries surrogate SCs [5-8]. Pseudocapacitive materials boast fast surface redox reactions and encompass high specific capacitance [9, 10]. Research is forged ahead to procure high specific capacitance from materials with pseudocapacitive properties from transition metal oxides like RuO<sub>2</sub> [11], Co<sub>3</sub>O<sub>4</sub> [12], NiO [13], CuO [14], SnO<sub>2</sub> and Manganese oxides [15]. The study explores the application of Mn<sub>3</sub>O<sub>4</sub> nanoparticles as electrodes for a supercapacitor to enhance its specific capacitance, highlighting the potential of the material when a transitional element is added to its composition [16, 17]. The material was chosen due to its high theoretical specific capacitance 1370 F/g [18], effective Jahn-Teller effect [18], wide potential window, variable oxidation states, environmental friendliness, non-toxicity, and cost-effectiveness [19, 20]. The survey indicates that the specific capacitance can reach up to 832 F/g [21] under different current densities and scan rates for thin films while it can reach 435 F/g at 1mV/s scan rate by hydrothermal method [22]. The study put in place that Ni-Al double-layered hydroxides had a specific capacitance value of 795 F/g at 1A/g, indicating good charge/discharge stability in solvothermal route [23]. To increase the specific capacitance of Mn<sub>3</sub>O<sub>4</sub> through hydrothermal method, transition element Nickel was doped at different concentrations of 0.3%, 0.6% and 0.9%. This work focuses on the electrochemical properties of Ni doped  $Mn_3O_4$  particles by hydrothermal way. Ni was chosen, due to its 0.72Å ionic radius, a close proximity to Mn's ionic radius of 0.80 Å, [24] resulting in minimal distortion in the structure and minimizing Jahn-Teller effect. The work explores and refines the electrochemical properties of the doped material, for its outwitting specific capacitance of 626 F/g at a specific current of 0.5 A/g for 0.6% Ni doped Mn<sub>3</sub>O<sub>4</sub>.

### **II.** Experimental Procedure

First  $Mn_3O_4$  nanoparticles lay the ground work and are characterized for elemental, microstructural, morphological and electrochemical properties. Then, Ni doped manganese oxide, Ni: $Mn_3O_4$  nanoparticles were synthesized for the betterment of electrochemical properties.

*Hydrothermal synthesis*: Hydrated Manganese Chloride (MnCl<sub>2</sub>.4H<sub>2</sub>O) and Sodium Hydroxide (NaOH) procured from Merck co., were taken as precursors [25]. 1M MnCl<sub>2</sub>.4H<sub>2</sub>O was primed by dissolving 4.95g of it, in 25ml of double distilled (DD) water in a 100ml beaker. 25ml of DD water was taken in a 50ml beaker and

1.0g of NaOH was dissolved in it for the composition of 1M NaOH solution. After the two solutes were completely dissolved, 100ml beaker containing Manganese Chloride was taken on the pan of a magnetic stirrer and was kept under stirring. Now, NaOH solution was added to  $Mn_3O_4$  solution at a snail's pace by drop wise, stirring  $Mn_3O_4$  solution continuously. Now the resultant liquid was transferred to Teflon lined autoclave. The autoclave was kept in a furnace and was heated for 6 hours at 450K. In this way the  $Mn_3O_4$  nano particles were prepared at three reaction time periods of 6hrs, 10hrs and 14hrs respectively at a reaction temperature of 450K. After the furnace was cooled to room temperature, autoclave was brought out from the furnace and the liquid in it was slowly filtered with filter paper lined to a funnel. It was washed with DD water and ethanol for few times. The precipitate in the filter paper was dried beneath halogen lamp for one hour. Further, these precipitate particles were grinded for one hour with a mortar. The obtained dark brown particles were the nanoparticles of  $Mn_3O_4$ . To prepare 0.3%, 0.6% and 0.9% Ni doped  $Mn_3O_4$  nanoparticles, 17mg, 34mg and 52mg of NiCl<sub>2</sub>.6H<sub>2</sub>O were added respectively to 4.95g  $MnCl_2.4H_2O$  in 25ml DD water.

*Material characterization:* X-Ray Diffraction (XRD), Raman spectroscopy, Scanning Electron Microscopy (SEM) and Electrochemical analysis manoeuvred for characterization of  $Mn_3O_4$  nanoparticles and Ni: $Mn_3O_4$  nanoparticles. Microstructure of the particles was delved by using X-ray diffractometer varying 2 $\theta$  in the range of 10° to 70° as displayed in figure 1. The Raman spectrum for the prepared nanoparticles was inspected in the wave number range of 200 to 900cm<sup>-1</sup>. Surface morphology was examined by scanning electron microscope. The elements unveiled in the nanoparticles were mounted from EDAX system. Electrochemical exploration was accomplished by composing electrodes from the prime powder, carbon black and polyvinylidene fluoride in 8:1:1 ratio. N-Methyl-2-Pyrrolidone (NMP) drops in few were added to the above grinded powder to make fine slurry. Chemically cleaned nickel strip substrate was brush coated with the slurry. NMP solvent was vaporized by heating to 480K. The prepared strip was working electrode in three electrode system for exploring Cyclic voltammetry (CV), Chronopotentiometry (CP) for galvanostatic charge discharge (GCD) studies and Electrochemical Impedance Spectrum (EIS) meant for the track down of specific capacitance.





Fig. 1: XRD spectra of Mn<sub>3</sub>O<sub>4</sub> nanoparticles synthesized at 6, 10, and 14 hours of reaction time periods at 450K temperature at 1M concentration of NaOH.

*XRD studies:* Fig. 1 uncovers XRD characteristics divulge by maximum diffraction peaks prop at 20 values of  $36.10^{\circ}$ ,  $32.35^{\circ}$ ,  $59.80^{\circ}$ ,  $28.95^{\circ}$ ,  $18.30^{\circ}$ ,  $58.45^{\circ}$ ,  $50.70^{\circ}$ ,  $44.40^{\circ}$  and  $64.60^{\circ}$ . They affirm the peak positions of JCPDS card no:89-4837 for Millar planes (211), (103), (224), (112), (101), (321), (105), (220) and (314) respectively. They promulgated I41/amd(141) space group owing to tetragonal structure for Mn<sub>3</sub>O<sub>4</sub>. It was in

consummation of the results obtained from the prior literatures reviewed [19]. From the primeval literature evaluation, reaction temperature was espoused at 450K. At 450K reaction temperature for the reaction time of 6 hours, the peaks designated to (211) and (103) rose to a little extent. At a reaction temperature of 450K, the reaction times were further increased to 10 and 14 hours. At a reaction time of 10 hours peaks consistent with (224) and (103) were also emerged along with heighten of antecedent peaks (211) and (103). When the reaction time was mounted up to 14 hours at the same temperature of 450K, particles manifest perpetual crystallinity with imperishable number of sharp peaks as put on display in fig. 1 and were borne for further processes. These optimal temperatures and time were in obedience to the quondam literatures of hydrothermal synthesis [26]. Lattice parameters evaluated from XRD data were a = b = 5.745Å and c = 9.346Å as witnessed in the Table no: 1. They were well harmonized with the standard JCPDS card no:89-4837. Parameters 'a' and 'b' were literal to graded values of JCPDS data, while 'c' deviated with a permittable percentage error 1.16.



**Fig. 2:** XRD spectra of Mn<sub>3</sub>O<sub>4</sub> nanoparticles synthesized with 0.5, 1.0 and 1.5M concentrations of NaOH at reaction temperature of 450K for 14 hours.

Table 1. Lattice parameters of Minso4 particles.				
S.No	Lattice parameter	Calculated In Å	Standard In Å	% Error
1	а	5.745	5.763	0.3
2	b	5.745	5.763	0.3
3	с	9.346	9.456	1.16

Table 1: Lattice parameters of Mn<sub>3</sub>O<sub>4</sub> particles.

The clout of variation of NaOH was also keenly detected. At 0.5M and 1.5M concentrations of NaOH, the spectral intensities were curtailed, indicating the best optimum concentration was 1M at locked reaction time of 14 hours at 450K. Also, with the increase of NaOH concentration, the characteristic peak (211) was slightly drifting towards lower values of 2 $\theta$ , which was the general trend with the increase of oxidizing agent. 1M NaOH concentration along with 14 hours reaction time at 450K was used for the further prepared Ni doped Mn<sub>3</sub>O<sub>4</sub> nanoparticles. The literature inspection, and the current experimental output have shown that Ni doping at the concentration levels taken up does not brought striking effect to the structure of Mn<sub>3</sub>O<sub>4</sub> [27] as the ionic radii of Mn and Ni were close. At 0.6% of Ni doping, the doped atoms occupied tetrahedral Ni<sup>2+</sup> ions occupied Mn<sup>2+</sup> tetrahedral sties to a maximum extent without occupying the same sites besides them which increased void space between these sites.

**Raman studies:** In the course of coalescence of the particles, the stoichiometric ratio of NaOH was rooted by suss out Raman spectra of the contrived particles. The superior peak at 654.55 cm<sup>-1</sup> devoted to stretching vibration symmetric mode  $A_{1g}$  of  $Mn^{2+}$  in Mn-O bond. The peak at 306.83 cm<sup>-1</sup> was pertained to out of plane bending of  $E_{g2}$  symmetric mode. And the peak at 360.66 cm<sup>-1</sup> was the sake of asymmetric stretch of bridge

oxygen structure Mn-O-Mn. With the hike of NaOH concentration from 0.5M to 1.5M through 1.0M, which was a precise stoichiometric ratio, the ascendant peak at 654 cm<sup>-1</sup> mounted maximum intensity. Furthermore, the peaks at 306.83 and 360.66 cm<sup>-1</sup> were also surge compared to other proportions of NaOH as shown in fig. 3. It was also chronic with XRD exploration. In XRD studies, slight drift of peaks was observed in fig. 2, which happens with the increase of oxidizing agent, NaOH.



Fig. 3: Raman spectra of Mn<sub>3</sub>O<sub>4</sub> particles synthesized at 450K for 14 hours, at 0.5, 1.0, and 1.5M NaOH concentrations.

The peaks at 307, 361cm<sup>-1</sup> were meagre and 654cm<sup>-1</sup> were low at reaction temperature of 450K for 14 hours for NaOH concentrations of 0.5M and 1.5M. For 1M NaOH concentration the characteristic peaks were more exhibitive indicating stoichiometric ratio at the same reaction temperature and time period as delineated in fig. 3. The doping of Ni atoms to the structure increased the stretching of M-O bonds both at tetrahedral and octahedral sites initially. When doping increases, with the occupancy of first tetrahedral sites, distortion occurred between the two tetrahedral sites. The distortion reduced when the doping was increased beyond 0.6%. Maximum stretching at this concentration created scope for the movement of ions for intercalation and deintercalation.

*Morphological studies:* To inquire the morphology of the particles, SEM exploration was taken up. Surface morphology of undoped and doped particles disseminates the particle morphology and was proliferating with the escalation of temperature. It was in concurrence with the results from literature consideration [28]. Figure 4 unfurl the SEM images (a), (b), (c) and (d), for pure  $Mn_3O_4$  nanoparticles at reaction times of 6, 10, 14 hours and 0.6% Ni: $Mn_3O_4$  particles at 14 hours reaction time, all synthesized at 450K. The structure in (d) was with more particle size when compared to previous three. It for bye ensured compact with the prior proclaimed literature [22]. Image (d) pictured in figure 4, perspicuously opens the particle size was dominant at 0.6% Ni, in comparison to other three. Further, setting (d) side by side with (c), the particle size in (d) for 0.6% was affluence little than that of in (c). EDAX spectrum of the  $Mn_3O_4$ , elementary overlay and EDAX spectrum of 0.6% Ni: $Mn_3O_4$  particles visualization confirm the phase pure composition of the particles as set out in figures 5, 6 and 7 respectively.



Fig. 4: SEM images of Mn<sub>3</sub>O<sub>4</sub> synthesized at 450K using for (a) 6hrs, (b) 10hrs, (c) 14hrs and (d) 14hrs for Ni: Mn<sub>3</sub>O<sub>4</sub> nanoparticles.



Fig. 5: EDAX spectrum for pure  $Mn_3O_4$  nanoparticles synthesized at 450K for 14hours at 1M NaOH concentration.



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# Fig. 6: EDAX images of Mn<sub>3</sub>O<sub>4</sub> nanoparticles, (a) Elementary overlay, APEX images of element (b) Oxygen and (c) Manganese.



Fig. 7: EDAX spectrum for  $Mn_3O_4$  nanoparticles doped with Ni at 0.6% concentration synthesized at 450K for 14 hours prepared at 1M NaOH concentration.

*Electrochemical analysis:* Working electrode was assembled from nanoparticulate sample powder, carbon black and Polyvinylidene fluoride (PVDF) quantified in the ratio 8:1:1, computed to 100 mg. Carbon black accomplish conduction while PVDF binds the particles. The pastiche was grinded for half an hour in a motor. Few drops of N-Methyl-2-Pyrrolidone (NMP) were added to it to make a fine homogeneous slurry. It was glazed on nickel plate with a brush, up to three quarters on one side. Owing the fact of 475 K boiling point for NMP, it was effaced by inflamed air in a furnace at 480K. Mn<sub>3</sub>O<sub>4</sub> and Ni:Mn<sub>3</sub>O<sub>4</sub> working electrodes were assembled by tightly encased copper wire to the other side of nickel plate and substrate was masked by Teflon

tape. CHI608C three-electrode electrochemical workstation was handled to explore the electrochemical properties of the electrodes.

*Cyclic voltametric studies:* Mn<sub>3</sub>O<sub>4</sub> working electrode, Platinum disc counter electrode and Platinum foil reference electrode in 1M Na<sub>2</sub>SO<sub>4</sub> electrolyte were maneuverer in a three-electrode electrochemical analyser. Super capacitive performance of the electrode material was evaluated from the CV curves inside the potential window of 1V, i.e., from -0.9V to 0.1V at scan rates of 3mV/s to 15mV/s in the steps of 3mV/s be seen in figure 7.  $\frac{\int I(V)dV}{2 \text{ m } S_r P_W}$  expression was employed for the reckoning the specific capacitance (C<sub>s</sub>) of the electrode material. Here  $\frac{1}{2} \int I(V) dV$  is area enclosed by CV loop, m is active mass, S<sub>r</sub> is scanning rate and P<sub>w</sub> is potential window. C<sub>s</sub> of Mn<sub>3</sub>O<sub>4</sub> electrodes quantified at 315, 291, 268, 251 and 232 F/g at scan rates of 3, 6, 9, 12 and 15mV/s respectively and were put on show in table 2. These values were in line with previous literature values of

hydrothermal process. In an idea to improvise them, another transition metal element in the form of nickel (Ni) was doped at various proportions. The doped Ni:Mn<sub>3</sub>O<sub>4</sub> electrodes at 0.3%, 0.6% and 0.9% of Ni were trailed for the furtherance of C<sub>s</sub>. The concerning C<sub>s</sub> at the scan rate of 3 mV/s emanated for 0.3%, 0.6% and 0.9% Ni:Mn<sub>3</sub>O<sub>4</sub> electrodes were 384, 421 and 396 F/g respectively. The C<sub>s</sub> values were increased by 21.9%, 33.7% and 25.7% by doping Ni at the said proportions in series compared with pure Mn<sub>3</sub>O<sub>4</sub> nanoparticles.

The augmentation in  $C_s$  values was ascribed to the increase in void space between side-by-side  $Mn^{2+}$  due to occupancy of one of it. changes in the ionic size and interaction between ions and electrons at the surface.  $Mn_3O_4$  structure consists of tetrahedral and octahedral elements for  $Mn^{2+}$  and  $Mn^{3+}$  ions respectively.  $Mn^{2+}$  ions are bonded to four oxygen ions and  $Mn^{3+}$  ions by six oxygen ions. Also packing fractions for these elements were 0.78 and 0.72 [28, 29]. Considering these factors, at initial doping percentages, Ni ions easily occupy  $Mn^{2+}$  sites until packing fraction decreases to its minimum.

The area of the CV loop was decreasing with the decrease in the scan rate. It means that, as the scan rate decreases, charges were finding time for intercalation and deintercalation. The area enclosed by the loop decreases to bring an ideal rectangular situation where the increase and decrease of voltage and current are almost parallel to voltage and current axes respectively. The area of the CV loop was low at 0.6% Ni concentration. It was more at concentrations below and above that concentration for a scan rate of 3mV/s. At scan rate below 3mV/s, the cures were fluctuating because of low current densities.



Fig. 8: CV curves of pure  $Mn_3O_4$  electrodes synthesized at 450K for 14 hours at scan rates of 3, 6, 9, 12 and 15mV/s.



S.No	Scan Rate in mV/s	Specific Capacitance in F/g
1	3	315
2	6	291
3	9	268
4	12	251
5	15	232

Table 2: Cs values from CV curves of Mn3O4 electrodes.

Table 3: Cs from CV curves of Ni:Mn3O4electrodes at scan rate of 3mV/s.

S.No	Ni concentration in %	Specific Capacitance at scan rate of 3mV/s in F/g
1	0.0	315
2	0.3	384
3	0.6	421
4	0.9	396

**Chrono potentiometric studies:** Galvanostatic charge discharge (GCD) curves were hassled for the  $Mn_3O_4$  and Ni: $Mn_3O_4$  electrodes as shown in figures 10, 11 and 12. Spinal structure of  $Mn_3O_4$  with  $Mn^{2+}$  ions in tetrahedral sites and  $Mn^{3+}$  in octahedral sites [26] with a formula of  $Mn^{2+}(Mn^{3+})_2O_4$  [30] give out high pseudocapacitance [31]. Specific capacitance was assessed from the formula,  $C_s = \frac{1 \Delta t}{m \Delta V}$ , where I is current through the electrode,  $\Delta t$  is discharging time and  $\Delta V$  is potential difference and m is active mass of the electrode. The figured specific capacitance for  $Mn_3O_4$  electrodes were 264, 302, 352, 454 and 565 F/g at specific currents of 1.50, 1.25, 1.00, 0.75 and 0.50 A/g respectively. All these values are shown in table 3. For Ni: $Mn_3O_4$  electrodes at 0.0%, 0.3%, 0.6% and 0.9% Ni concentrations, the specific capacitance were 352, 426, 476 and 414 F/g respectively at a current density of 1A/g. These values were expose to view in table 4. The value of specific capacitance was maximum for 0.6% Ni doped Ni: $Mn_3O_4$  electrodes. The swing in these values also confirm the trends in the values drawn from CV curves. Finally, 0.6% Ni doped  $Mn_3O_4$  electrodes were ply to study the GCD curves and its specific capacitance was found at specific currents of 1.50, 1.25, 1.00, 0.75 and 0.50A/g as 429, 448, 476, 529 and 626 F/g respectively. In the modus operandi, extortionate  $C_s$  cherished through hydrothermal course of action from  $Mn_3O_4$  after doped with Ni.

S.No	Current density in A/g	Specific Capacitance in F/g
1	1.50	264
2	1.25	302
3	1.00	352
4	0.75	445
5	0.50	565

Table 4: C<sub>s</sub> values of Mn<sub>3</sub>O<sub>4</sub> electrodes form GCD curves.







Fig. 11: CP curves of Ni:Mn<sub>3</sub>O<sub>4</sub> electrodes.

S.No	% Ni concentration	Specific Capacitance at 1A/g in F/g
1	0.0 (Pure Mn <sub>3</sub> O <sub>4</sub> )	352
2	0.3	426
3	0.6	476
4	0.9	414

Table 5: C <sub>s</sub>	values of Ni:Mn <sub>3</sub> O <sub>4</sub>	electrodes form	GCD curves.
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Fig. 12: CP curves of 0.6% Ni doped Mn<sub>3</sub>O<sub>4</sub> electrodes.

Specific capacitance values from the aforementioned literature skimmed were observed and compared with the values obtained from the present work for the particles synthesized by hydrothermal method. Z. Liu et al synthesized hollow microcubes of  $Mn_3O_4$  which have specific capacitance of 176 F/g at a specific current of 0.3 A/g [30]. Dadamiah P.M.D. Shaik et al synthesized octahedron shaped  $Mn_3O_4$  octahedrons having specific capacitance of 198 F/g at 0.5mA/cm<sup>2</sup> [32]. H. Ullah Shah produce nanoparticles of  $Mn_3O_4$  with a specific capacitance value 355 F/g at a specific current of 0.35 A/g [33]. Ultra fine nanoparticles of  $Mn_3O_4$  exhibited a value of 401 F/g at a scan rate of 10mV/s [34]. Porous nanoparticles of  $Mn_3O_4$  were produced by Dadamiah PMD Shaik exhibited specific capacitance value of 435 F/g at a scan rate of 1 mV/s [35]. All these utterances were shown in table 7 precisely.

S.No	Specific current in A/g	Specific Capacitance in F/g
1	1.50	429
2	1.25	448
3	1.00	476
4	0.75	529
5	0.50	626

Table 6: C<sub>s</sub> values of 0.6%Ni:Mn<sub>3</sub>O<sub>4</sub> electrodes form GCD curves.

*Electrochemical Impedance Spectroscopic studies:* To substantially confirm the capacitive nature and cyclic stability of the 0.6% Ni:Mn<sub>3</sub>O<sub>4</sub> electrodes, EIS studies were accompanied on these electrodes. Ran through 3600 cycles, the solution resistance was increased from  $12\Omega$  to  $27\Omega$ , charge transfer resistance was decreased from  $203\Omega$  to  $104\Omega$  and the specific capacitance was found to be 82%. With the decrease of charge transfer resistance from 203 to  $104 \Omega$ , conductance increases in the same proportion, which leads to the curtailment of capacitance. It was also noticed in the decline of time period, when time period was set side by side between 1<sup>st</sup> cycle and  $3600^{th}$  cycle as shown in fig. 12. For these electrodes there was a low decrease of specific capacitance even after 3600 cycles.



Fig. 13: Nyquist plots of 6% Ni doped Mn<sub>3</sub>O<sub>4</sub> nanoparticles.



Fig. 14: Change of Specific Capacitance with cycle number.

#### V. Conclusion

Hydrothermal method of synthesis was followed for the preparations of  $Mn_3O_4$ , Ni: $Mn_3O_4$ nanoparticles. Crystalline structure was seasoned from XRD and Raman spectroscopic studies. At a reaction temperature 500K and reaction time of 15hours reliable yield resulted in the formation of decent crystalline nanoparticles of Ni: $Mn_3O_4$ . Porous structure was rooted from morphology captured from scanning electron microscopy. Ni doped  $Mn_3O_4$  nanoparticles with the doping concentration of 0.6% Ni exhibited better porous nature which is the key for super capacitors. Super capacitive properties of the Ni: $Mn_3O_4$  electrodes have shown a specific capacitance of 626 F/g at specific current of 0.50 A/g for 6% Ni concentration. These electrodes have exhibited a very good cyclic stability of 82% of the specific capacitance even after 3600 cycles. Finally, by hydrothermal method, maximum C<sub>s</sub> value was successfully achieved from  $Mn_3O_4$ , by doping with Ni.

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