# Investigation of The Relationship between Groups and Subgroups in C3S's Structure Transition

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**Abstract:** In recent years, many studies have been carried out on cement and its phase's tounderstand the morphology, and to control the mineralogy of this material; due to the greatposition it has become globally occupied. This material is formed from a synthetic rock calledclinker; tricalcium silicate (Ca3SiO5 or C3S) its major constituent present a concentrationfrom 40% to 70%, and its solid solution with various impurities is called alite. Impure C3Sexhibits seven polymorphs from ambient temperature to 1500°C; three forms triclinic (T1, T2, T3), three monoclinic (M1, M2, M3) and one shape rhombohedral (R). At room temperature, impurities stabilize some of the high temperature forms of the pure compound. Those forms are related by transformation matrix determined in this article. The aim of the present paper isto investigate the structural modulations of alite.

*Key words*: Cement, clinker Portland, Polymorph, crystal structure, alite, space group, sub-group, matrix and coordinates system.

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# I. Introduction

Alite is the most important phase in cement Portland, it forms 60 to 70% of clinkerPortland, and it is a solid solution of C3S, modified by substitution or insertion of foreign ionsin its crystal structure. In a high temperature, the thermal agitation factor of atomic positionwill increase and with the effect of atomic interaction, some atoms live their position, and leta hole, which will be occupied by another atom came from impurities which has the same size of the lifted one; those substitution and/or insertion may be the key of structure transitions. In kiln, using a combination of differential thermal analysis (ATD), optic microscopy and X-ray diffraction (XRD), several transformations were observed of alite polymorphism [1].



It is usually found in clinkers in a monoclinic form (M1, M3 or a mixture of both), and rarely in T2 shape[2]. Among those structures, just 4 polymorphs have a structural model, which are T1, T2, M3 and R; Recently De Noirfontaine has presented M1 form based on group and sub-group relationship[3].

The first study of alite was done by Jeffery [4]on a single crystal, he showed that the threepolymorphs T1, R and M3 have a similar structure, and he gives a pseudo one common forthe true of the three forms, this structure was confirmed by O'Daniel on single crystal sample. The cell parameters with hexagonal axis ( $a_{H}=b_{H}=7\text{\AA}$  &  $c_{H}=25\text{\AA}$ ) and space group (R3m)were determined for the composition of alite: 54(CaO).16(SiO<sub>2</sub>).Al<sub>2</sub>O<sub>3</sub>.MgO with Z=9formulas by cell (81 atom per cell). The tree structures are very close; they are different by thecalcium ions and oxygen's atoms of SiO<sub>4</sub> tetrahedra, they are different by the disorder oforientation in tetrahedra SiO<sub>4</sub><sup>4-</sup>.

# **II.** Space group transformations

Nishi and Takéuchi were solved at the same way the rhombohedral R[5] and the monoclinicM3[6]polymorphs, this M3 structure characterized by a big number of atoms (228 per cell,table 1) with different and complicated disordered orientation of  $SiO_4$  tetrahedra. In 1995,Mumme determined the substructure of M3 polymorph, similar to that presented by Nishiwith an average structure

 $M3_{Nishi} = <3 \times 1 \times 2 > M3_{Mumme}$ (table 1)[7]. In 2012, using the structural relationship between triclinic T1 and monoclinic M3 polymorphs, M1 was solved by DeNoirfontaine and a structural model was given in table 1[2], [3]. In 2005, Maria Angeles De LaTorre modified Nishi structure or M3 with reducing the number of atomic position based ondisorder orientation[8].

The triclinic shape was studied by Golovastikov for T1 and he proposed a P-1 space group[9]as mentioned in De Noirfontaine thesis[2]. In 2004, Peterson gives the structure of T2 based onGolovastikov model and using Synchrotron powder diffraction, the cell metrics are noted intable 1[10]. In 2008, De La Torre has been solved the structure T3 by combined the X-ray powder diffraction (XRPD), Neutron powder diffraction (NPD) and Synchrotron powder diffraction (SPD) using Rietveld method[1].

Since 1952, uncountable studies are worked on cement and clinker's phases, tounderstand the reactions of their constituents and the arrangement of atomic positions, but thestructure of the main phase of clinker which is alite is still misunderstood. A structural presentation will be given to explain the transitions between all the models of C3S structure.

### 2.1 Rhombohedra supergroup

All trigonal space groups with secondary or tertiary symmetry elements havemonoclinic C-centered maximal translationengleiche subgroups of index [3]. The figure 1,samurais the relationship between all polymorphs found in literature; R3m space group isconsidered as the super-group of Cm (space group of M3 polymorph), which is related to thesub-groups Pc and Pm (space groups of M1 polymorph) with the same index (figure 2). Alltrigonal space groups without secondary or tertiary symmetry elements have triclinic maximal subgroups of index [3]. In this study we will try to give the matrix transformation of R3m toCm and from Cm to Pc, Pm and to P1 which is the Golovastikov model.



Figure 1: Subgroups Graph of rhombohedra system [11]

To start we have first to identify the index, and how to calculate it. The index is the factor by which the number of symmetry operations has been reduced; it is the number of symmetricelements which are not changed passing from the group G to the subgroup H with the matrix fransformation (P, p). It is equal to the number of stable symmetric element in G devise by the symmetric elements of the sub-group H.

$$[i] = \frac{order \quad of \quad group \quad G}{order \quad of \quad subgroup \quad H} = \frac{number \quad of \quad position \quad in \quad G}{number \quad of \quad position \quad in \quad H} \times \frac{1}{Z}$$
$$[i] = \frac{number \quad of \quad symetric \quad element \quad in \quad G}{number \quad of \quad symetric \quad element \quad in \quad H}$$

The number of general positions equivalent for a space group, is the number of positions generated by Bravais letter multiply by the number of position corresponding to crystallineclass.

#### *Number of general position= Bravais letter* × *Class number*

For R3m group, we have a primitive lattice presents (with rhombohedral axis) rotation axisorder 3 along [111], gives 3 positions and the mirror gives 2, so we have six different generalpositions are:(x, y, z) (-y, x-y, z) (-x+y, -x, z) (-x+y, y, z) and (x, x-y, z):

And for Rhombohedra system which are (in obverse R) tree sites of symmetry are : (0,0,0);(2/3,1/3,1/3); (1/3,2/3,2/3).

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**Figure 2: Monoclinic transformations** 

From Jeffery and Nishi model, The rhombohedral structure R3m has the multiplicityZ=9, so for alite (Ca3SiO5) we have 72 crystallographically independent sites: 18 Ca atomsall in general position [site  $3\times9b$ ]; 9 Si atoms in special position SP (00z) [site of  $3\times3a$ ]; and 45 Oxygen atom: 6 in special position (00z) [site  $6\times3a$ ] and 3 in general position [ $3\times9b$ ](Figure 4).



Figure 3: Rhombohedral Hexagonal transformation in ITC [12]

# 2.2 Rhombohedral-monoclinic transformation

There are two main considered transformations in phase transition and group-subgrouprelationship, are Transformation of the coordinate system and symmetry operations. The matrix of transformation is written as follow:

$$\begin{bmatrix} a & b & c \end{bmatrix}_{R} = \begin{bmatrix} a & b & c \end{bmatrix} \times P(1)$$
$$\begin{bmatrix} x \\ y \\ z \end{bmatrix}_{R} = P^{-1} \times \begin{bmatrix} x \\ y \\ z \end{bmatrix} (2)$$

To study the transformation from R3m to Cm, we have first to change the basis of hombohedral system to hexagonal axis to facilitate the calculation. According to figure 3 and figure 5a, the transformation can be written as:

$$a_{R} = \frac{2}{3}a_{H} + \frac{1}{3}b_{H} + \frac{1}{3}c_{H}$$

$$b_{R} = \frac{-1}{3}a_{H} + \frac{1}{3}b_{H} + \frac{1}{3}c_{H}$$

$$c_{R} = \frac{-1}{3}a_{H} - \frac{2}{3}b_{H} + \frac{1}{3}c_{H}$$

$$[a \ b \ c]_{R} = [a \ b \ c]_{H} \times P(3)$$

$$\begin{bmatrix} 2}{3} & \frac{-1}{3} & \frac{-1}{3} \\ \frac{1}{3} & \frac{1}{3} & \frac{-2}{3} \\ \frac{1}{3} & \frac{1}{3} & \frac{1}{3} \end{bmatrix}$$

With

We can calculate even the inverse transformation Q using the identity matrix by:

$$\Rightarrow Q = P^{-1} = \begin{bmatrix} 1 & 0 & 1 \\ -1 & 1 & 1 \\ 0 & -1 & 1 \end{bmatrix} (5)$$

 $P = \begin{bmatrix} \frac{2}{3} & \frac{-1}{3} & \frac{-1}{3} \\ \frac{1}{3} & \frac{1}{3} & \frac{-2}{3} \\ \frac{1}{3} & \frac{1}{3} & \frac{1}{3} \end{bmatrix} (4)$ 

Equations (3) and (5) give the matrix of transformation P from the old basis, where the inverse matrix Q transforms the coordinates. So we can say that after defining the matrix P of rhombohedral ---> hexagonal transformation, calculation or finding all parameters information became easier. For atomic position, we have the Equation (6) below[11]:

$$\begin{bmatrix} x \\ y \\ z \end{bmatrix}_{R} = \begin{bmatrix} x \\ y \\ z \end{bmatrix}_{H} \times Q = \begin{bmatrix} x \\ y \\ z \end{bmatrix}_{H} \times \begin{bmatrix} 1 & 0 & 1 \\ -1 & 1 & 1 \\ 0 & -1 & 1 \end{bmatrix} (6)$$

So based on those calculations, we can calculate or find metric unit cell (equation 3), millerindices (Equation 7) and atomic positions (equation 6), just from the matrix oftransformation[11]:

$$\begin{bmatrix} h & k & l \end{bmatrix}_{R} = \begin{bmatrix} h & k & l \end{bmatrix}_{H} \times P$$

$$\begin{bmatrix} h & k & l \end{bmatrix}_{R} = \begin{bmatrix} h & k & l \end{bmatrix}_{H} \times \begin{bmatrix} \frac{2}{3} & \frac{-1}{3} & \frac{-1}{3} \\ \frac{1}{3} & \frac{1}{3} & \frac{-2}{3} \\ \frac{1}{3} & \frac{1}{3} & \frac{1}{3} \end{bmatrix} (7)$$

The same as before for hexagonal-Monoclinic transformation, all matrixes are confirmed by ITC after calculated;



Fig. 5.1.3.10. Rhombohedral lattice with primitive rhombohedral cell a, b, c and the three centred monoclinic cells. (a) *C*-centred cells  $C_1$  with  $a_1, b_1, c'; C_2$  with  $a_2, b_2, c';$  and  $C_3$  with  $a_3, b_3, c'$ . The unique monoclinic axes are  $b_1, b_2$  and  $b_3$ , respectively. Origin for all four cells is the same. (b) *A*-centred cells  $A_1$  with  $a', b_1, c_1; A_2$  with  $a', b_2, c_2;$  and  $A_3$  with  $a', b_3, c_3$ . The unique monoclinic axes are  $c_1, c_2$  and  $c_3$ , **Figure 4:Rhombohedral monoclinic transformation in ITC** [12]

From the figure 4 we can write:

$$a_H = \frac{3}{2}a_M - \frac{1}{2}b_M + c_M$$
$$b_H = b_M$$
$$c_H = c_M$$

So :

$$\begin{bmatrix} a & b & c \end{bmatrix}_{H} = \begin{bmatrix} a & b & c \end{bmatrix}_{M} \times P'$$
$$\begin{bmatrix} a & b & c \end{bmatrix}_{H} = \begin{bmatrix} a & b & c \end{bmatrix}_{M} \times \begin{bmatrix} \frac{3}{2} & 0 & 0 \\ -\frac{1}{2} & 1 & 0 \\ 1 & 0 & 1 \end{bmatrix} (8)$$

In this way from equation (3) and (8) we can calculate:

$$\begin{bmatrix} a & b & c \end{bmatrix}_{R} = \begin{bmatrix} a & b & c \end{bmatrix}_{H} \times P = \begin{bmatrix} a & b & c \end{bmatrix}_{M} \times P' \times P$$

$$\begin{bmatrix} a & b & c \end{bmatrix}_{R} = \begin{bmatrix} a & b & c \end{bmatrix}_{M} \times \begin{bmatrix} \frac{3}{2} & 0 & 0 \\ -\frac{1}{2} & 1 & 0 \\ 1 & 0 & 1 \end{bmatrix} \times \begin{bmatrix} \frac{2}{3} & -\frac{1}{3} & -\frac{1}{3} \\ \frac{1}{3} & \frac{1}{3} & -\frac{2}{3} \\ \frac{1}{3} & \frac{1}{3} & \frac{1}{3} \end{bmatrix}$$
(9)

These calculations give the matrix of Rhombohedra- monoclinic transformation P".

$$\begin{bmatrix} a & b & c \end{bmatrix}_{R} = \begin{bmatrix} a & b & c \end{bmatrix}_{M} \times \begin{bmatrix} 1 & -\frac{1}{2} & -\frac{1}{2} \\ 0 & \frac{1}{2} & -\frac{1}{2} \\ 1 & 0 & 0 \end{bmatrix} (10)$$



Figure 5: Unit cell parameters transformation

Those matrixes are confirmed by International Tables of Crystallography VA[12] for the choice1 in obverse setting (figure 6);

Primitive rhombohedral cell $\rightarrow$ triple hexagonal cell $R_1$ , obverse setting (Fig. 5.1.3.6c)	$\begin{pmatrix} 1 & 0 & 1 \\ \overline{I} & 1 & 1 \\ 0 & \overline{I} & 1 \end{pmatrix}$	4 = 4 = 4 = 4 = 4 = 4 = 4 = 4 = 4 = 4 =	Rhombohedral space groups (cf. Section 4.3.5)
Triple hexagonal cell $R$ , obverse setting $\rightarrow$ C-centred monoclinic cell, unique axis <b>b</b> , cell choice 1 (Fig. 5.1.3.9 <i>a</i> ) <b>c</b> and <b>b</b> axes invariant	$\begin{pmatrix} \frac{2}{3} & 0 & 0\\ \frac{1}{3} & 1 & 0\\ \frac{7}{3} & 0 & 1 \end{pmatrix}$	$\begin{pmatrix} \frac{3}{2} & 0 & 0\\ \frac{7}{2} & 1 & 0\\ 1 & 0 & 1 \end{pmatrix}$	Rhombohedral space groups (cf. Section 4,3.5)
Primitive rhombohedral cell $\rightarrow$ C-centred monoclinic cell, unique axis <b>b</b> , cell choice 1 (Fig. 5.1.3.10 <i>a</i> ) [111] <sub>r</sub> $\rightarrow$ <b>c</b> <sub>m</sub>	$\begin{pmatrix} 0 & 0 & 1 \\ \overline{1} & 1 & 1 \\ \overline{1} & \overline{1} & 1 \end{pmatrix}$	$\begin{pmatrix} 1 & \frac{1}{2} & \frac{1}{2} \\ 0 & \frac{1}{2} & \frac{1}{2} \\ 1 & 0 & 0 \end{pmatrix}$	Rhombohedral space groups (cf. Section 4.3.5)

Figure 6: Selected 3 × 3 transformation matrices P and Q =P-1[12]

Taking the graph below (Figure 7), the C1m1 is considered as non-isomorphic subgroup of R3m which is a special kind of klassengleiche subgroups (subgroup belongs to the samecrystal class; it has lost translation symmetry; its primitive cell larger than that of G) and belong to the same space group type (Hermann-Mauguin symbol) [12]; type "I", it preserve the position 1 and number 4 plus those obtained by centering translation.

Gen	erato	rs sel	ected (	E); r(1	,0,0);	1(0,1,0	)); <i>1</i> (	(0,0,1); /	( <del>1</del> , <del>1</del> ,	÷); (2); (4)				
Posi	tions													
Mahir	plicity	ŝ.			6	loordin	ates					Reflection	n conditions	
Wyck Site s	off let ymmet	ter, try		(0,0	$ 0\rangle$	(†,†,†)	)+	(i,i,i)+				General:		
18	e l	I	(1) x.y. (4) <i>g</i> , <i>x</i> .	2 1	(2) ý. (5) <i>X</i>	x = y,z + y <sub>c</sub> y,z		(3) x+y (6) x,x+	,Ē,Ţ			λλίζ : - λλ <u>ί</u> Ο : - λδΩλί : Ι λδοτ : λ 000/ : Ι λδο0 : λ	-h + k + l = 3n -h + k = 3n = 3n + l = 3n = 3n a = 3n	
												Special: 1	no extra conditions	
9	b	m	1.1.1		x.2x.z	2 22	18. <i>8.</i>	ž						
1	119	Law	0.0					-						
Sym Alon a' = Origi	metr g  001  (2a -  n at 0	y of s   p31   b) (0,z	pecial pr "b" = +{	ojecti –a+t	005 i)		Ali a' s Ori	ong [100] / = j( <b>a</b> + 2 <b>b</b> igin at x,0	91 ) 0	$b'=\frac{1}{2}(-a-2b$	+ ¢)	Along [210 $\mathbf{a}' = \frac{1}{2}\mathbf{b}$ Origin at x,	p l w l b = +c $\frac{1}{2}x, 0$	
Max 1 Ila Ilb	imal [2] ([3]) [3]) [2])	<b>nom-i</b> <u>R314</u> <u>R1m(</u> R1m( R1m) R1m( R3m1 R3c(a	somorpl 23, 146) C m, 8) C m, 8) C m, 8) (156) ( = -a, b	iic sul (1: - (1: - (1; 1) (1; 1) 1; 2 = −b	bgroup 2; 3)+ 4)+ 5)+ 6)+ ; 3; 4; ,c'=2	5: 6 c)(161)								
Max	imal	isom	orphic si	ihgro	ups of	lowest	ind	ex						
IIc	[2]	R3m(	a' = -a, t	=-	h,c' = 2	c) (160	); [4]	R3m(a'=	= -2	a, b' = -2b)(160)	0			
Mini 1 11	imal [2], [3],	non-i R3m( P31m	somorph 166); [4] / (a' = }[2	ic sup #43m a + b)	ergros (215);   ,b' = }(	nps  4  F 4 3  -a + b	lur (2 ), e' :	216): [4] <i>1</i> = <del>]</del> c) (157	13.m	(217)				
			Fig	ire	7: F	Rhor	րհ	ohed	ral	space gr	oup in T	TC [12]	1	
			8'							- space St	osp m i	-0[12	1	

Monoclinic system for C3S has 3 forms, M1, M2, and M3; the known structures are forM3 and M1, we will study one form which is M3 superstructure of Angeles De La Torre[8], themultiplicity is equal to 36, so it found:

M3-Ca3SiO57  $\rightarrow$  the monoclinic structural description has 155 crystallographicallyindependent sites: 54Calcium atoms [18 atom in special position (x0z) and 36 in general position]; 18 Silicon atoms all are in special position; and 83 Oxygen atoms [56 atoms inspecial position (x0z) and 27 in general positions].

2.3

The graph thereafter (figure 8), and the figure 2, shows the relation between Cm (M3 space group) and Pc, the space group of M1 superstructure proposed by De Noirfontaine [2].

First, before talking about matrix transformation, we need to take a look on ITC for Cm space group (Figure 8):

CONTINUED		No. 8	Cm
Generators selected (7); a)	0.01 +0.1.01 +0.0.11 +(1.1.0) (2)		
Positions Materia es	Camiliato	Reflection	militions
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4 4 1 (liaga	(2) 3.5.2	Adf : h + k AOf : h - k O(f) : h - 2 O(f) : h - 2 h(0) : h + k O(0) : k - 3 AO(0) : h - 3	= 2a = 2a
1000 DA		Special an	extra anniditions
Symmetry of special projecti Along [001] $< 11m$ $\pi = \pi$ , $b' = b$ Origin at 0.0.2	ants. Along [100] $p$ [ $w$ ] $a^{*} = [b  b^{*} - c_{u}]$ Origin at $z, 0, 0$	Along $ 010  \neq i$ $a^{i} = c$ $b^{i} =  a $ Origin at $0, y, 0$	
Maximal non-isomorphic online           I         [2] C1 (P3, 1)         1+           Hz         [2] P1 a1 (Px, 7)         [, 2]           [2] P1 a1 (Px, 6)         [, 2]           [2] P1 a1 (Px, 6)         [, 2]           [3] P1 a1 (Px, 6)         [, 2]	hgraupa + (1, 1, 0) - (2) / 1 с 1 м' = 20 м° с, 9)		
Maximal isomorphic subground for a state of the subground	ups of lowest index a + 2c, c' = 2c) (Cm, R; [1] C 1 m 1 (H = 3)	\$6(Cm.3)	
Minimal non-isomorphic sup I [2] C2/w(12); [2] Cmir2 [3] P3ir1 (156); [3] P3 to	ergroops (35x [2] Cmc 2, (36x [2] Amm2 (36x [2] n (157); [3] #3m (160)	Aem2(39); [2]Fmm2(42); [2]Imm2(4	4); [7] <i>1 ma</i> 2 (46);
II $  2  P1 = 1 \le a = \frac{1}{2}a, b = \frac{1}{2}b$	k) (Pm, 40		

# Figure 8: Monoclinic space group in ITCA [12]

The present section gives a brief summary, of the sub- and supergroup data in the space group Cm; the select subgroup P1a1 (conventional group is P1c1) has been classed in non-isomorphic subgroup of C1m1; type II which is a klassengleiche transformation or called "k subgroup"; that means that the transformation is accompanied with preserve of the same crystalline class and losing all centric translation; and the letter "a" means that C1m1 and P1c1 have the same conventional cells. We keep the positions: (x y z) and (x+1/2, -y+1/2, z) (Figure 8).

$C_i^3$	C1m1		No. 8	Cm
UNIQUE AXIS b, CELL	THORE I			
Generators selected (1).	$(1,0,0)$ ; $r(0,1,0)$ ; $r(0,0,1)$ ; $r(\frac{1}{2},\frac{1}{2})$	0); (7)		
General position				
Mahiplicity.		Coordina	tes-	
Wychoff letter, Site Aynumetry		1,0,0)+ (	1,0)+	
4 8 1		(I) s.y.t (2	1 x.J.z	
I Maximal translationengle	che saligroups			
[2] C1 (1.Pl)	1+		t/2(a - b), t/2(a + b)	e
II Maximal Manangleiche	abgroups			
· Low of centring transla	fiam			
121 Plat (7, Pici)	$1; 2 + (\frac{1}{2}, \frac{1}{2}, 0)$		-a-c.b.a	0.1/4.0
17) Plan1 (6)				

Figure 9: Monoclinic space group in ITC A1[13]

Figure 9, present the same space group Cm in International tables for crystallography volume A1 [13], this edition of tables are specify for symmetry relations between space groups, from this figure and figure 5b we can extract the relation between P1a1 and C1m1, where we have:

$$a_{Pc} = -a_{Cm} - c_{Cm}$$
$$b_{Pc} = b_{Cm} \qquad (14)$$
$$c_{Pc} = a_{Cm}$$

So the matrix P here and the inverse Q could be written as:

$$P = \begin{bmatrix} -1 & 0 & 1 \\ 0 & 1 & 0 \\ -1 & 0 & 0 \end{bmatrix} \Rightarrow Q = P^{-1} = \begin{bmatrix} 0 & 0 & -1 \\ 0 & 1 & 0 \\ 1 & 0 & -1 \end{bmatrix}$$
(15)

For general transformation of coordinate system, both the basis and the origin have to betransformed; with p, which is the translation matrix or coordinate system of the new origin O'in the old one[14].

$$P = \begin{bmatrix} -1 & 0 & 1 \\ 0 & 1 & 0 \\ -1 & 0 & 0 \end{bmatrix} \Rightarrow p = \begin{bmatrix} 0 \\ 1/4 \\ 0 \end{bmatrix}$$
$$\begin{bmatrix} P, p \end{bmatrix} = \begin{bmatrix} -1 & 0 & 1 & 0 \\ 0 & 1 & 0 & 1/4 \\ -1 & 0 & 0 & 0 \end{bmatrix} (16)$$
$$p' = -P^{-1} \times p = \begin{bmatrix} 0 & 0 & -1 \\ 0 & 1 & 0 \\ 1 & 0 & 1 \end{bmatrix} \times \begin{bmatrix} 0 \\ 1/4 \\ 0 \\ 0 \end{bmatrix} (17)$$

p' corresponds to the position of the old origin in the new coordinate system[15]. Combined with the origin shift, the new coordinates result from the old ones according to:

$$\begin{bmatrix} x \\ y \\ z \end{bmatrix}_{P_{C}} = \begin{bmatrix} P^{-1}, p' \end{bmatrix} \times \begin{bmatrix} x \\ y \\ z \end{bmatrix}_{Cm}$$
$$\begin{bmatrix} x \\ y \\ z \end{bmatrix}_{P_{C}} = \begin{bmatrix} 0 & 0 & -1 & 0 \\ 0 & 1 & 0 & -\frac{1}{4} \\ 1 & 0 & 1 & 0 \end{bmatrix} \times \begin{bmatrix} x \\ y \\ z \end{bmatrix}_{Cm} (19)$$

So from equation (14) and (18) we find the atomic position and unit cell parameterstransformed from Cm to Pc space group.

$$x_{Pc} = z_{Cm}$$
$$y_{Pc} = y_{Cm} - \frac{1}{4}$$
$$z_{Pc} = x_{Cm} - z_{Cm}$$

Those calculations are explained in ITC Vol A [12] and using the same calculations of rhombohedralmonoclinic transformation, we can confirm the result mentioned above infigure 10 and figure 11:

Cell choice $1 \rightarrow \text{cell choice } 2 \{P \rightarrow P\}$			
$1 \subset \neg A$ Cell choice $2 \rightarrow \text{cell choice } is \begin{cases} P \rightarrow P \\ A \rightarrow I \end{cases}$ Unique axis is invariant	$\begin{pmatrix} 1 & 0 & 1 \\ 0 & 1 & 0 \\ 1 & 0 & 0 \end{pmatrix}$	$\begin{pmatrix} 0 & 0 & 1 \\ 0 & t & 0 \\ 1 & 0 & 1 \end{pmatrix}$	Mountlinic (cf. Section 2.2.16)
Coll choice $3 \rightarrow \text{ortf. choice } 1 : \begin{cases} P \rightarrow P \\ I \rightarrow C \end{cases}$	20.027.201	3.5 5 67	
(Fig. 3.1.3.2a)			-
Cell choice $1 \rightarrow cell$ choice $2 \begin{cases} P \rightarrow P \\ A \rightarrow R \end{cases}$	2012/22	12/17/3	
Cell choice $2 \rightarrow cell$ choice $3 \begin{cases} P \rightarrow P & Unique axis e \\ R \rightarrow J & invariant \end{cases}$	$\begin{pmatrix} 0 & 1 & 0 \\ 1 & 1 & 0 \end{pmatrix}$	(1)	Monoclass: (cf. Section 2.2.16)
Cell choice $3 \rightarrow cell choice 1$ $\begin{pmatrix} P \rightarrow P \\ I = 3 \end{pmatrix}$	0 0 1/	0 0 1/	Contract of the second second
(Fig. 53,3.26)			
Cell choice $1 - mil$ choice $2 \begin{cases} P \rightarrow P \\ R \rightarrow C \end{cases}$	0.000		
Cell choice $2 \rightarrow \text{cell choice } $ $\begin{cases} P \rightarrow P & \text{Usight ation a} \\ C \rightarrow t & \text{invariant} \end{cases}$		$\begin{pmatrix} 1 & 0 & 0 \\ 0 & 1 & 1 \end{pmatrix}$	Monoclinic (cf. Section 2.2.10)
Cell choice $3 \rightarrow and choice 1$ $\begin{cases} P \rightarrow P \\ I \rightarrow B \end{cases}$	10 1 17	(0 1 0)	A
(Fig. 5.1.3.2c)			



Figure 11: Monoclinic transformation equivalent[12]

#### 2.4 Triclinic structure

To find the metric and positions relationship between known polymorphs is not easy,especially for triclinic structure which is characterized by its low symmetry, it has one axisorder 1 rotated in the inverse direction; the triclinic structure which is likely found incommercial clinker is the shape T2 [13]; this structure was solved by Peterson who gives 83crystallographic sites:29 Calcium atom [25 atoms in general position and 4 in special position( $0\ 0\ 0$ ); ( $0\ 2\ 0$ ); ( $1\ 2\ 0\ 2$ ) and ( $1\ 2\ 2\ 2\ 2$ ); 9 Silicon atom in general positions; and 45 Oxygenall are occupied general positions.

And for structural description of De La Torre [1]model for T3 form, he has the same 83crystallographically independent sites: 29 Calcium [4 in special position (x0z) and 25 ingeneral position]; 9 Silicon all in general position; and 45 Oxygen in general position.

CONTINUED	No. 2	P1

Generators selected (1): t(1,0,0); t(0,1,0); t(0,0,1); (2)

Positions Multiplicity, Wyckoff letter,		Coordinates	Reflection conditions
Site symmetry			General:
2 1 1	(1) x,y,z	(2) <i>x</i> , <i>y</i> , <i>z</i>	no-conditions
			Special: no extra condition

Figure 12: Triclinic space group in ITC[12]

### III. Conclusions

This article is a bibliographic review on the polymorphism of C3S; it discusses the matrixtransformations based on group sub-groups relationship. This work reports a study of theorystallographic structure of alite polymorphs and involves information concerning themechanism of transitions between all forms. The calculation and determination of matrixtransformation in C3S transitions will facilitate solving structures. This work will besupported after; by another article whish applied those matrixes to find atomic positions ofalite in industrial samples.

#### References

- [1] Á. G. De la Torre, R. N. De Vera, A. J. M. Cuberos, and M. A. G. Aranda, "Crystal structure of low magnesium-content alite:
- Application to Rietveld quantitative phase analysis," *Cem. Concr. Res.*, vol. 38, no. 11, pp. 1261–1269, 2008.
   M.-N. de Noirfontaine, "Étude Structurale Et Cristallographie Du Composé Majoritaire Du Ciment Anhydre : Le Silicate
- Tricalcique," École Polytechnique, 2000.
- [3] M. N. De Noirfontaine, M. Courtial, M. De Noirfontaine, F. Dunstetter, G. Gasecki, and M. Signes-Frehel, "Tricalcium silicate Ca3SiO5 superstructure analysis: a route towards the structure of the M1 polymorph," *Zeitschrift für Krist.*, vol. 227, no. 2, pp. 102–112, 2012.
- [4] J. W. Jeffery, "The crystal structure of tricalcium silicate," Acta Crystallogr., vol. 5, no. 1, pp. 26–35, 1952.
- [5] F. Nishi and Y. Takeuchi, "The rhombohedral structure of tricalcium silicate at 1200 °C," Zeitschrift fur Krist., vol. 168, no. 1–4, pp. 197–212, 1984.
- [6] F. Nishi and Y. Takéuchi, "Tricalcium silicate Ca30[SiO4]: The monoclinic superstructure," Zeitschrift für Krist., vol. 172, pp. 297–314, 1985.
- [7] W. G. Mumme, "Crystal Structure of Tricalcium Silicate from a Portland Cement Clinker and its Application to Quantitative XRD Analysis," *Neues Jahrb. Fur Mineral.*, vol. 4, no. 4, pp. 145–160, 1995.
- [8] A. G. De La Torre; S. Bruque; J. Campo; M. A. G. Aranda, "The superstructure of C3S from synchrotron and neutron powder

- diffraction and its role in quantitative phase analyses," *Cem. Concr. Res.*, vol. 32, pp. 1347–1356, 2002. N. I. Golovastikov, R. G. Matveeva, and N. V Belov, "Crystal Structure of the Tricalcium Silicate 3CaO.SiO2 = C3S," *Sov.* [9] physics, Crystallogr., vol. 4, no. 441, p. 20, 1975.
- V. K. Peterson, B. a Hunter, and A. Ray, "Tricalcium Silicate T1 and T2 Polymorphic Investigations: Rietveld Refinement at [10] Various Temperatures Using Synchrotron Powder Diffraction," J. Am. Ceram. Soc., vol. 87, no. 9, pp. 1625–1634, 2004.
- U. Müller, Symmetry Relationships between Crystal Structures. Oxford University Press, 2013. [11]
- [12] [13] T. Hahn, International tables for crystallography Volume A, Fifth edit. the international union of crystallography, 2002.
- U. M. H. Wondratschek, International tables for crystallography Volume A1, First edit. Boston, London: Kluwer Academic Publishers, 2004.
- [14] H. F. W. Taylor, Cement chemistry, 2nd editio., vol. 20, no. 4. London: Thomas Telford Publishing, 1997.
- [15] J. M. P.-M. and M. I. A. S. Ivantchev, E. Kroumova, G. Madariaga, "Bilbao Crystallographic Serve," Group-Subgroup Lattice and Chains of Maximal Subgroups. Departamentos de Física de la Materia Condensada y Física Aplicada II, Universidad del País Vasco, Spain, 2000.

Tableau 1: Cell parameters of known polymorphs of C3S

Form	Chemical composition	Space group	a (Å)	b (Å)	c (Å)	α (°)	β (°)	γ (°)	Z	V (Å <sup>3</sup> )	References
	Ca <sub>2.99</sub> Na <sub>0.01</sub> (Si <sub>0.90</sub> Al <sub>0.04</sub> Fe <sub>0.02</sub> P <sub>0.03</sub> Mg <sub>0.05</sub> )O <sub>5</sub>	Cm	12,235	7,073	9,298	06	116,31	06	9	721.3	[7]
M3	Ca <sub>2,89</sub> Mg <sub>0,11</sub> (S iO <sub>4</sub> )O	Cm	33,083	7,027	18,499	06	94,1	06	36	4289	[9]
	Ca <sub>54</sub> MgSi <sub>16</sub> Al <sub>2</sub> O 90	Cm	33,1078	7,0355	18,5211	06	94,137	06	36	4302.9	[8]
M1	r	Pc	27.87438	7.0591	12.2578	06	116.031	06	18	2167	[8]
X	Ca <sub>2,98</sub> Si <sub>0,98</sub> Al <sub>0,</sub> 04O5	R3m	7,135	7,135	25,586	06	06	120	6	1128	[5]
	Ca <sub>3</sub> (SiO <sub>4</sub> )O	R3m	L	L	25	06	06	120	6	1060.9	[4]
T1	Ca <sub>3</sub> (SiO4)O	$P^{-1}$	11,67	14,24	13,72	105,5	94,3	06	18	2190	[6]
T2	Ca <sub>3</sub> SiO <sub>5</sub>	$\mathbf{P}^{\mathrm{I}}$	11.7416	14.2785	13.7732	105.129	94.415	89.889	18	2222.02	[10]
Т3	Ca <sub>2.96</sub> Mg <sub>0.05</sub> Al <sub>0.01</sub> ( Si <sub>0.99</sub> Al <sub>0.01</sub> )O <sub>5</sub>	$\mathrm{P}^{-1}$	11.6389	14.1716	13.6434	104.982	94.622	90.107	18	2166.228	[1]